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RESEARCH NOTE

Effect of TiO₂ Nanofiber Density on Organic-Inorganic Based Hybrid Solar Cells

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ABSTRACT

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1. INTRODUCTION

Low-cost semiconductor materials and facile fabrication routes for photovoltaic junctions have been the longstanding goals of photovoltaic materials research. New materials or fabrication procedures which substantially reduce the cost of photovoltaic electricity, can help driving a rapid expansion in the implementation of photovoltaic technology [1]. Although the polymer-fullerene solar cells have obtained 7.4% efficiency, the use of fullerene electron acceptors suffers from poor photo stability and low electron mobility through phase segregation. Replacing the fullerene with inorganic semiconductor nanocrystals as electron acceptors and transport components is an alternative approach, leading to hybrid solar cells. This replacement has the advantages of superior physical and chemical stability, higher electron mobility, facile fabrication property, the ability to engineer interfacial band offsets, and consequently the photovoltage [2-8]. Organic hybrid

In this work, a comparative study of hybrid solar cells based on P3HT and TiO₂ nanofibers was accomplished. Electrospinning, a low cost production method for large area nanofibrous films, was employed to fabricate the organic-inorganic hybrid solar cells based on poly (3-hexylthiophene) and TiO₂ nanofibers. The performance of the hybrid solar cells was analyzed for four density levels of TiO₂ nanofibers which resulted in the average power conversion efficiency of about 0.0134% under AM 1.5 simulated illuminations (100 mWcm⁻²). It is found that the higher densities of TiO₂ lead to more interface area and generating exciton, so the power conversion efficiency will be increased till the active layer thickness is about 400 nm; but, with increasing thickness more than 400 nm the increasing density did not have positive effect.

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solar cells based on semiconductor nanocrystals are among the most promising alternatives to traditional silicon solar cells [2, 9]. Hybrid solar cells have received widespread attention for cost-effective manufacturing, and therefore have the potential to be applied in consumer products because they are thin, light-weight and flexible. Also, these have the capability of low-temperature processing and the simplicity of produce in large-area [1-3, 7, 10-12].

The basic principles of organic–inorganic hybrid solar cells are similar to the bulk heterojunction concept, utilized by p-type conjugated polymers and ntype inorganic semiconductors. Hybrid thin film solar cells based on poly-hexylthiophene (P3HT) and cadmium sulfide (CdS) [13], cadmium selenide (CdSe) [14], titanium oxide (TiO₂) [15], zinc oxide (ZnO) [16], copper indium disulfide (CuInS₂) [17], copper indium diselenide (CuInSe₂) [18] in the forms of nanoparticle, nanowire and nanorod have been developed as the next generation cells. Titanium oxide (TiO₂) has attracted great research interest since it has already been successfully used in dye sensitized solar cells [19]. It has been found that ultrafast photo induced charge transfer could occur at the TiO₂-

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conjugated polymer interface [20, 21]. Typically, two common approaches have been used to fabricate TiO_2 -polymer solar cells. One is to infiltrate conjugated polymer into nanostructured TiO_2 film [22, 23] and the other one is the simultaneous deposition of the blend of polymer and TiO_2 nanocrystals [24, 25].

In the case of the porous TiO₂-polymer system, the current power conversion efficiency (PCE) is still low, although many efforts have been made towards the optimization of the structure of titania network and the architecture of the device [23, 26, 27]. Actually, the fabrication of nano-structured TiO₂ film with controlled pore size and pore distribution together with the complete infiltration of polymer into this network has still been a challenge so far. For the blend of polymer and TiO₂ nanocrystals, the poor electric contact among the TiO₂ particles and the cathode electrode is one serious problem. Therefore, the well aligned one dimensional (1D) TiO₂ arrays that enable a large interfacial area and continuous carrier pathway could be the ideal structure for the hybrid solar cell. However, the growth of TiO₂ nanowires with précised density and size control, and high aspect ratio remains a significant technological challenge [26].

Electrospinning has been developed to be a simple and versatile method to prepare several of continuous 1D fibers, including polymers, ceramics, composites and metals, with diameters down to several nanometers [23]. After the first demonstration of electrospun TiO₂ nanofibers [28], many efforts have been done toward their application in photovoltaic devices since the electrospun nanofibers can be fabricated with controllable size, density and orientation.

More importantly, electrospinning is a low cost production method for large area nanofibrous films. Since area and cost are two important issues for solar cells, electrospinning is a technical idea for the fabrication of hybrid solar cells. In this paper, we studied the hybrid solar cells based on electrospun TiO_2 nanofibers and regioregular P3HT. The effects of TiO_2 nanofibers density and active layer thickness have been surveyed as well.

2. MATERIALS AND METHODS

2. 1. Materials Glass substrates with FTO electrode were prepared from Solaronix. Titanium tetra isopropoxide ($C_{12}H_{28}O_4Ti$, 97%), Titanium tetrabutoxide (Ti(OCH₂CH₂CH₂CH₃)₄, 97%), polyvinylpyrolidone ((C_6H_9NO)X, M_w 1300000) and Poly 3-hexylthiophene (($C_{10}H_{14}S$)_n, M_w/M_n<2.5, M_w : 54000-75000) were purchased from Sigma Aldrich. Acetic acid (CH₃COOH, 100%) and ethanol (C_2H_5OH , 99.9%) were supplied from Merk.



Figure 1. Flowchart of fabrication process

2. 2. Device Fabrication The general process of fabrication has been shown in Figure 1 and more details explained in resumption. As substrates, glass sheets of 1×1 cm² covered with indium tin oxide (ITO) and fluorine doped tin oxide (FTO) were used with a sheet resistance of 8 Ω / cm².

The TCO was patterned by etching with mixture of HCl and Zn powder for several minutes. The part of the substrate that forms the contact is covered with Scotch tape, preventing etching. The scotch tape was removed after etching and the substrate cleaned by a routine procedure, which included sonication in detergent followed by repeated rinsing in distilled water, acetone, and isopropanol, in sequence. Cleaned TCO glass is first coated with compact TiO₂ layer by spin coating the sol gel solution (0.25M titanium butoxide in ethanol) at 6000 rpm [6]. The compact TiO₂ layer had some cracks. To fill these cracks, spin coating the sol gel was repeated 4 times. The thickness of each layer was 60 nm. Then, TiO₂ nanofibrous film was prepared on top of the compact layer. The sol gel for spinning nanofibers contained Titanium tetraisopropoxide (0.56 g), acetic acid (0.81 gr), ethanol (2 gr) and PVP (0.76 g). The sol gel was stirred 15 minutes. Sol gel was electrospun onto substrates at a rate of 2 mL/h under an electric field of 20 kV. It was spun in four different densities by controlling the volume of injected sol gel, and then sintered at 500°C for 1 hour by rapid thermal processing to remove organic components and allow nucleation and growth of TiO₂ grains. The thickness of the sintered TiO₂ nanofibrous film was about 200 nm, which is the optimum thickness for the hybrid solar cells [28]. A

layer of P3HT was infiltrated in each TiO_2 nanofibrous film by spin-coating P3HT solution (regioregularity is 50 mg/mL in dichlorobenzene) at 1500 rpm for 30 s and subsequent thermal annealing at 100°C for 20 min in a glove box filled with high purity nitrogen. Then, 40 nm gold electrode is evaporated on top of the film through a shadow mask by using a thermal evaporator. Figure 2 shows the schematic structure and energy level diagram of the hybrid solar cell.

2. 3. Characterization The TiO₂ compact layer and active layer prepared on glass substrates were used for profilometry. Profilometry was performed on DEKTAK3 instrument. Also, the TiO_2 nanofibers prepared by electrospinning on glass substrates were used for the purpose of XRD and spectroscopic characterizations. The XRD operation was carried out on a Bruker D8-Discover instrument. Ultraviolet-vis (UV-vis) absorption spectra were recorded on a HR 4000 CG-UV-NIR spectrophotometer (wavelength range 100-1000 nm). The morphology and structure of the nanofibers were characterized using scanning electron microscopy (SEM, HITACHI S41-60) with an accelerating voltage of 10 kV. Performance of photovoltaic cells was measured using a calibrated AM 1.5 solar simulator under white light with intensity of 100 mW/cm^2 .



b

Figure 2. a) Schematic structure of a $TiO_2/P3HT$ hybrid solar cell, b) energy level diagram



Figure 3. SEM images of TiO₂ nanofibers in diffrent density

3. RESULTS AND DISCUSSIONS

We introduced TiO₂ nanofibers as the n-type inorganic semiconductor and P3HT as the p-type organic semiconducting polymer in the organic-inorganic hybrid solar cell. The morphology of TiO₂ nanofibers is shown in Figure 3 as an SEM image. We could observe the average diameter of nanofibers was in the range of 60– 100. As Figure 3 shows, four densities of nanofibers from low (a) to high (d) have been fabricated for analyzing the effect of the density of nanofiber on cell's performance. These different densities has been made by controlling the volume of sol gel injected in the electrospinning system. With increasing density, the thickness of active layer will grow too.

Figure 4 demonstrates the diffractogram of TiO_2 nanofibers which are collected on an FTO substrate. Also, the corresponding anatase and rutile patterns are included for comparison. On an FTO substrate, the bare TiO_2 shows a mixture of anatase/rutile phases. The crystallographic compositions are calculated using the following equation [29]:

Content of anatase =
$$P_A/(1.265*P_R+P_A)$$
 (1)

where, P_A and P_R are the main peak intensities of anatase and rutile crystalline phases, respectively. The achieved TiO₂ phase contains 53% anatase and 47% rutile. Zhang et al., have compared the electrical properties of rutile and anatase TiO₂ and proved that, while TiO₂ in its anatase phase is highly conductive, it is highly resistive in its rutile crystalline structure [28]. Higher amount of O_{vac} can be formed in anatase in comparison with rutile. This is the reason why anatase crystalline phase shows higher conductivity and better photocatalytic properties. The optical density of various layers of the solar cell is shown in Figure 5. The preliminary P3HT exhibits abroad absorption spectrum peaking at about 520 nm and another absorption peaks at around 300 nm. With TiO_2 nanofibers, the optical density of P3HT: TiO_2 active layer was slightly increased. This result shows that the incorporation of TiO_2 nanofibers into P3HT does not lead to degrade the optical quality of the composite films.



Figure 4. XRD patterns of T_1O_2 nanotibers in maximum density



Figure 5. UV-vis absorbance spectra of the solar cell with P3HT:TiO₂ active layer.



Figure 6. Current density–voltage curves for TiO_2 :P3HT nanocomposite films. Spinning sol gel in volume: a) 0.025 ml b) 0.05 ml c) 0.075 ml d) 0.1 ml

TABLE 1. Characteristic parameters of cell with structure FTO/TiO₂/TiO₂:P3HT/Au under AM 1.5 illumination.

	active layer thickness (nm)	J _{sc} (mA/cm ²)	V _{oc} (V)	FF	η (%)
a	293	0.092	0.25	0.21	0.0049
b	360	0.125	0.27	0.24	0.0081
c	412	0.179	0.23	0.32	0.0134
d	475	0.121	0.24	0.22	0.0064

Corresponding to the diameter of nanofiber, the thickness of the inorganic layer was between around 300 to 500 nm. We obtained the photovoltaic performances of P3HT with TiO₂ nanofibers in four fiber densities as shown in Table 1 and Figure 6. The enhanced current density in P3HT composite films with TiO₂ nanofibers was obtained by incorporation of enough amounts of TiO₂ nanofibers. This is achieved by an effective electron transfer via n-type semiconducting TiO₂ nanofibers to anode through increasing the density and the effective area between the organic and inorganic phase in the active layer to generate more electron hole from excitons [1].

Since, exciton generation occurs in the interface of two phases, so more density of fibers in a constant thickness (lower than 400 nm) is ideal. The main limiting factor is suboptimum nanostructure morphology; which it causes too little of the polymer volume lies within an exciton diffusion length between two adjacent fibers. Thus, fibers with lower diameter form denser and thinner layer with more interfaces than squatty fibers. The values of J_{SC} , V_{OC} , FF and power conversion efficiency are presented in Table 1. The power conversion efficiencies of the devices are remained in low values in this work in which P3HT has been used as an active layer of the organic solar cell. However, the device performances such as J_{SC}, V_{OC} and FF can be slightly enhanced with the modification of TiO₂ nanofibers with dye. However, the device performances such as JSC, VOC and FF can be slightly enhanced with the modification of TiO_2 nanofibers with dye. In addition, the diameter of TiO₂ fiber and its crystallinity will affect the performance, which can be analyzed in future researches.

4. CONCLUSION

In this study, TiO₂ nanofibers and P3HT composite layer has been fabricated as an active layer for solar cells. Also, the optimized thickness of TiO₂ nanofibers layer was clarified by the SEM image which has been obtained in the range of 300-400 nm. We can state that the roles of TiO₂ nanofibers into P3HT are much significant, since the fibers make active layers and prepare more specific area. For improving performance, we will analyze the effect of N719 dye as a modifier and PEDOT:PSS layer in future investigations.

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Keywords: Hybrid Solar Cell TiO2 Nanofiber Density Power Conversion Efficiency Exciton در این تحقیق، بررسی مقایسهای روی سلول خورشیدی هیبریدی ساخته شده بر پایه پلی مر P3HT و دیاکسیدتیتانیوم انجام شده است. برای تولید فیلم نانوفیبر در سطح وسیع به منظور ساخت سلول خورشیدی آلی-معدنی بر پایه P3HT و نانوفیبر دیاکسیدتیتانیوم از یک روش کم هزینه به نام الکتروریسی استفاده شده است. عملکرد سلول های هیبریدی خورشیدی در چهار سطح تراکم نانوفیبر دیاکسیدتیتانیوم بررسی شده است که نتیجه آن راندمان تبدیل توان ۱۰۱۳۶ درصد در شرایط تابشی 1.5 AM میباشد. سلولهای ساخته شده با تراکمهای بیشتر نانوفیبر دیاکسیدتیتانیوم به دلیل دارا بودن سطح تماس بیشتر دو فاز و درنتیجه تولید اکسایتون بیشتر، راندمان تبدیل توان با افزایش تراکم تا جایی است که ضخامت لایه فعال حدود ۲۰۰ نانومتر باشد. با افزایش ضخامت لایه فعال به بیش از ۲۰۰ نانومتر افزایش تراکم اثر مثبت نخواهد داشت.

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